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# Nano Fe<sub>3</sub>O<sub>4</sub>-Activated Carbon Composites for Aqueous Supercapacitors (Nanokomposit Fe<sub>3</sub>O<sub>4</sub>-Karbon Aktif untuk Super-Kapasitor Akueus)

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### ABSTRACT

In this study, a symmetric supercapacitor has been fabricated by adopting the nanostructured iron oxide  $(Fe_3O_4)$ -activated carbon (AC) composite as the core electrode materials. The composite electrodes were prepared via a facile mechanical mixing process and PTFE polymeric solution has been used as the electrode material binder. Structural analysis of the nanocomposite electrodes were characterized by scanning electron microscopy (SEM) and Brunauer-Emmett-Teller (BET) analysis. The electrochemical performances of the prepared supercapacitor were studied using cyclic voltammetry (CV) and electrochemical impedance spectroscopy (EIS) in 1.0 M Na<sub>2</sub>SO<sub>3</sub> and 1.0 M Na<sub>2</sub>SO<sub>4</sub> aqueous solutions, respectively. The experimental results showed that the highest specific capacitance of 43 F/g is achieved with a fairly low  $Fe_3O_4$  has improved the capacitive performance of the composite via pseudocapacitance charge storage mechanism as well as the enhancement on the specific surface areas of the electrode. However, further increasing of the  $Fe_3O_4$  content in the electrode is found to distort the capacitive performance and deteriorate the specific surface area of the electrode, mainly due to the aggregation of the  $Fe_3O_4$  particles within the composite. Additionally, the CV results showed that the  $Fe_3O_4$  content is a better charge storage performance if compared with  $Na_2SO_4$  solution. It is believed that  $Fe_3O_4$  nanoparticles can provide favourable surface adsorption sites for sulphite (SO<sub>3</sub><sup>2-</sup>) anions which act as catalysts for subsequent redox and intercalation reactions.

Keywords: Activated carbon; aqueous electrolyte; iron oxide; supercapacitor

# ABSTRAK

Dalam kajian ini, super-kapasitor bersimetri telah dibina dengan menggunakan nanokomposit ferum oksida ( $Fe_3O_4$ )karbon aktif (AC) sebagai bahan asas elektrod. Elektrod nanokomposit telah disediakan secara percampuran mekanikal mudah dan larutan polimer PTFE telah digunakan sebagai agen pengikat untuk bahan elektrod. Pencirian struktur elektrod nanokomposit telah dilakukan dengan mikroskopi pengimbasan elektron (SEM) dan analisa Brunauer-Emmett-Teller (BET). Pencapaian elektro-kimia untuk super-kapasitor dalam larutan akueus 1.0 M Na<sub>2</sub>SO<sub>3</sub> dan 1.0 M Na<sub>2</sub>SO<sub>4</sub> telah dianalisis dengan mengunakan voltametri siklik (CV) dan spektroskopi impedansi elektro-kimia (EIS). Keputusan kajian menunjukkan kemuatan kapasitan tertinggi sebanyak 43 F/g boleh dicapai dengan penambahan kecil nanozarah  $Fe_3O_4$  dalam larutan 1 M Na<sub>2</sub>SO<sub>3</sub>. Ini jelas menunjukkan penambahan nanozarah  $Fe_3O_4$ pada kandungan yang rendah dapat meningkatkan pencapaian kemuatan kapasitan elektrod komposit menerusi mekanisme caj penyimpanan pseudokapasitan dan juga menambahkan keluasan permukaan spesifik elektrod. Walau bagaimanapun, penambahan kandungan  $Fe_3O_4$  yang lebih tinggi di dalam elektrod didapati akan menjejaskan pencapaian kapasitan dan mengurangkan keluasan permukaan spesifik elektrod, disebabkan penggumpalan nanozarah  $Fe_3O_4$  di dalam komposit. Tambahan pula, keputusan CV menunjukkan pencapaian caj penyimpanan elektrod nano-komposit  $Fe_3O_4$  di dalam kepasitan tapak penjerapan anion sulfida (SO<sub>3</sub><sup>2-</sup>) yang bertindak sebagai pemangkin untuk tindak balas interkalasi dan redoks seterusnya.

Kata kunci: Elektrolit akueus; karbon aktif; nanozarah ferum oksida; super-kapasitor

# INTRODUCTION

Global warming, air-pollution emission, rising oil prices and fossil fuel depletion have received critical attention in the new global economy in which a progressive swing from an oil-based economy to an electricity-based civilization is observed (Pasquier et al. 2003). Transforming natural energy from sustainable and renewable resources such as the sun, ocean tides and wind could yield high electricity energy output. However, man has limited control over the natural phenomena due to the climate/geographical constraints. For these reasons, energy storage technology appears as one of the most promising options in tackling the problem since renewably-generated electrical energy can be stored during the optimum period for future use.

Electrochemical capacitor, also known as supercapacitor, is a unique energy storage device designed for high power density, high charge-discharge cycle life and high discharge efficiency (Wu 2002). It differs from conventional capacitor by storing charge using the double layer concept in which charge accumulation occurs at the interface between electrode and electrolyte via the double-layer charging effects. The additional contribution of pseudo capacitance may also be part of the observed capacitance due to some participation of functional groups present on the electrode surface in redox reactions or chemisorption processes involving partial charge transfer (Conway et al. 2007).

On the basis of electrode materials used and the charge storage mechanisms, the core electrode material of supercapacitor generally can be classified into three main categories: Carbon materials with high surface area, such as activated carbon and carbon aerogel; transition metal oxides such as RuO<sub>2</sub>, NiO<sub>2</sub>, MnO<sub>2</sub>, Fe<sub>3</sub>O<sub>4</sub> and conducting polymers such as polyaniline and polypyrrole (Kalpana et al. 2006; Wang et al. 2009; Zhang & Chen 2008). The carbon systems employ the double layer storage mechanism involving the charge separation at the interface between the carbon electrode and electrolyte. No charge transfer takes place across the interface of the electrode material. For this reason, this process is non-Faradaic in nature and therefore we classify the carbonaceous materials as electric double layer capacitor or EDLC (Cottineau et al. 2005). In contrast, the charge storage mechanism achieved by the transition metal oxides and conducting polymers is Faradaic in nature. The redox reaction between electrode materials and electrolyte ions in the appropriate voltage window gives rise to the pseudo-capacitance through the electron transfer that is produced from the oxidation state changes, according to Faraday's laws (Shukla et al. 2000).

Over the past few years, significant effort has been devoted to the fabrication of advanced nanostructured metal oxides as the core electrode material, especially on hydrous RuO<sub>2</sub>.xH<sub>2</sub>O (Hu & Chen 2004; Hu et al. 2007; Kim & Popov 2002). RuO, is widely studied because it is conductive and has three accessible distinct oxidation states. The pseudo-capacitive behaviour of RuO, in acidic solutions has been the focus of research in the past 30 years (Simon & Gogotsi 2008). Although RuO<sub>2</sub>.nH<sub>2</sub>O gives high specific capacitance, its application has been hindered by its high cost (Du et al. 2009). Therefore, there is a need to search for less expensive pseudocapacitance materials to support the growing demand on supercapacitor in manufacturing industry, as a result of rapid advancement of consumer electronic products and technology. A number of research studies have shown an increasing interest in other transition metal oxides with various oxidation states, such as NiO (Ganesh et al. 2006; Nathan et al. 2008; Yeong et al. 2005; Yuan et al. 2005), MnO (Belanger et al. 2008; Lee & Goodenough 1999; Nagarajan et al. 2006), SnO, (Jayalakshimi et al. 2007; Kuo & Wu 2003; Lu et al. 2010), Fe<sub>2</sub>O<sub>3</sub> (Mallouki et al. 2006; Nagarajan & Zhitomirsky 2006; Xie et al. 2011) and Fe<sub>3</sub>O<sub>4</sub> (Brousse & Belanger 2003; Chen 2009; Wu et al. 2003).

Iron oxides have been identified as the potential electrode materials in view of the low-cost and

environmentally friendly nature. Recently, large capacitance exhibited by iron powder electrodes has been reported in alkali sulphite and sulfate solutions (Brousse & Belanger 2003; Chen 2009; Dubal et al. 2012; Nagarajan & Zhirtomirkey 2006; Sassin et al. 2010; Wang et al. 2008, 2006; Xie et al. 2011). Wu (2002) firstly reported that Fe<sub>3</sub>O<sub>4</sub>-SnO<sub>2</sub> composite electrode can achieve a specific capacitance of 33 F/g in 1.0 M Na<sub>2</sub>SO<sub>4</sub> aqueous solution. It is suggested that the low conductivity of Fe<sub>3</sub>O<sub>4</sub> combined with other conductive materials such as SnO<sub>2</sub> is essential to obtain large capacitance. In order to identify the effect of different electrolytes on Fe<sub>2</sub>O<sub>4</sub> electrodes, Wu et al. (2003) also have prepared an electrode that is consisted of a mixture of Fe<sub>2</sub>O<sub>4</sub> (90 wt. %), CB (10 wt. %), PTFE (4 wt. %) via coprecipitation method. The specific capacitances of the composite electrode are found to be 38, 5.3, 4.9, 5.8 F/g in 1 M Na<sub>2</sub>SO<sub>3</sub>, 1 M Na<sub>2</sub>SO<sub>4</sub>, 1M NaCl and 1M KOH respectively.

The findings of these two studies (Wu 2002; Wu et al. 2003) are consistent with another investigation on capacitive mechanism of Fe<sub>2</sub>O<sub>4</sub> in Na<sub>2</sub>SO<sub>2</sub>, Na<sub>2</sub>SO<sub>4</sub> and KOH aqueous (Wang et al. 2006) and suggested that the capacitance of the oxide was found to be sensitive to the anion species. The experimental results indicated that  $Fe_3O_4$  electrodes in Na<sub>2</sub>SO<sub>3</sub> exhibit the highest specific capacitance value (170 F/g), followed by  $Na_2SO_4$  (25 F/g) and KOH (3 F/g). In the work of Du et al. (2009), they reported the capacitive characteristic of nanostructured  $Fe_2O_4$ -AC based asymmetric supercapacitor in 6 M KOH solution. This hybrid system (a positive  $Fe_2O_4$  electrode and a negative AC electrode) is found to deliver a specific capacitance of 37.9 F/g within potential range of 0-1.2 V. A composite supercapacitor containing Fe<sub>3</sub>O<sub>4</sub> nanoparticles dispersed in multiwalled carbon nanotubes (MWNTs) was reported by Kim and Park (2011). The Fe<sub>2</sub>O<sub>4</sub>/MWNTs composite electrode exhibits specific capacitance of 165 F/g in 1 M Na<sub>2</sub>SO<sub>3</sub>, which this value is almost 3-fold enhancement if compared to the pure MWNT electrode (58 F/g).

To our best knowledge, there have been few investigations on  $Fe_3O_4/AC$  nanocomposite supercapacitor to date. In this work, we explore the nanostructured  $Fe_3O_4/AC$  composite as the core electrode materials via simple facile mechanical mixing method for the first time. Although the poor electric conductivity of  $Fe_3O_4$  limits its application in high power storage devices, the incorporation of carbon black can improve the electrochemical performance of the  $Fe_3O_4$  loading composite electrodes.

## MATERIALS AND METHODS

#### PREPARATION OF ELECTRODES

Active materials were made by doping different amount of  $\text{Fe}_3\text{O}_4$  nanopowder (Sigma Aldrich, particle size < 50 nm, BET surface ~ 38 m<sup>2</sup>g<sup>-1</sup>) to AC nanopowder (Sigma Aldrich, BET surface ~ 900 m<sup>2</sup>g<sup>-1</sup>) to form composite electrodes, containing carbon black powder (Ketjenblack, EC-600JD from Akzo Nobel, surface area ~ 1400 m<sup>2</sup>/g) and polytetrafluoroethylene (PTFE) in the weight ratio of 80:15:5. The mixture was dissolved into a 1:1 mixed solution of distilled water and isopropanol. Fifteen wt. % carbon black was added to increase the electrode conductivity of Fe<sub>3</sub>O<sub>4</sub> nanopowder (Du et al. 2009). The mixtures in dough form were then cold rolled into 0.8 mm thick films. Electrodes were punched into coin-shaped electrodes of 2 cm<sup>2</sup> and were dried in a vacuum oven at 80°C for 45 min to ensure the isopropanol solvent was evaporated completely. After being immersed overnight in 1 M Na<sub>2</sub>SO<sub>3</sub> and 1 M Na<sub>2</sub>SO<sub>4</sub>, respectively, the discs were then mounted onto a stainless steel-current collector in a Teflon cell holder by means of screw-fitting plug for electrochemical measurements.

### STRUCTURAL ANALYSIS

The surface morphology of the composite was observed using a field-emission scanning electron microscope (FESEM, FEI, Quanta 400 F). Five-point surface area measurement using Brunauer-Emmett-Teller (BET) was made with a Micromeritic, ASAP 2020 surface area analyzer under  $N_2$  gas.

### ELECTROCHEMICAL CHARACTERIZATION

Electrochemical tests including cyclic voltammogram (CV) and electrochemical impedance spectroscopy (EIS) were performed using Nova software in Autolab potentiostat (model: PGSTAT128N). In this study, all electrodes were tested in 1 M Na<sub>2</sub>SO<sub>3</sub> and 1 M Na<sub>2</sub>SO<sub>4</sub> solutions that were prepared by dissolving the appropriate amount of Na<sub>2</sub>SO<sub>3</sub> and Na<sub>2</sub>SO<sub>4</sub> salts, respectively, in deionized water. The respective specific capacitances (C<sub>s</sub>) were determined by using half integrated area of CV curve to obtain the charge (Q) and subsequently dividing the charge by the mass of the electrode (*m*) and the width of the potential window ( $\Delta$ V) following equation: C<sub>s</sub> = Q/m  $\Delta$ V.

#### **RESULTS AND DISCUSSION**

# MICROSTRUCTURAL CHARACTERIZATION

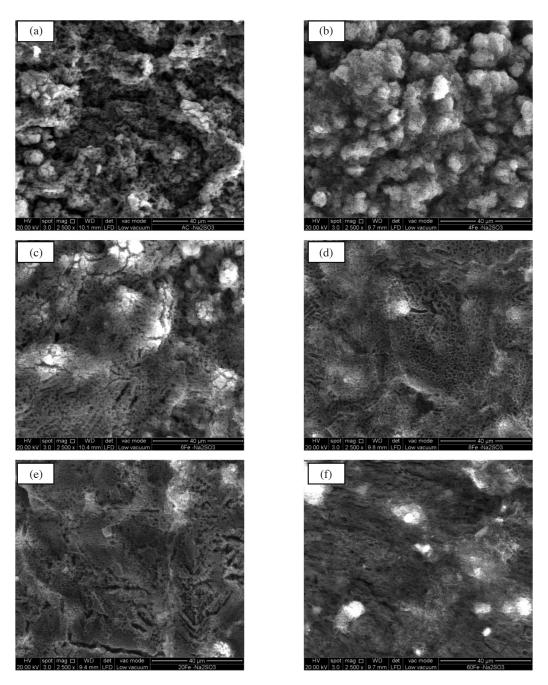
Scanning electron micrographs in Figure 1(a) to 1(f) indicates different amount of iron oxide loaded onto the composite electrodes. SEM photograph 1(a) and 1(b) show that the pitted and fragmented surfaces are more prominent for the pure AC and 4 wt. % Fe<sub>3</sub>O<sub>4</sub> composite electrodes while the formation of compact and rigid surface structures are observed for samples with increasing Fe<sub>3</sub>O<sub>4</sub> loading, as depicted in SEM photograph 1(e) and 1(f). SEM photograph 1(a) and 1(b) show that the pitted and fragmented surfaces are more prominent for the pure AC and 4 wt. % Fe<sub>3</sub>O<sub>4</sub> composite electrodes while the formation of compact and rigid surface structures are more prominent for the pure AC and 4 wt. % Fe<sub>3</sub>O<sub>4</sub> composite electrodes while the formation of compact and rigid surface structures are observed for samples with increasing Fe<sub>3</sub>O<sub>4</sub> loading, as depicted in SEM photograph 1 (a) and 1 (b) show that the pitted and fragmented surfaces are more prominent for the pure AC and 4 wt. % Fe<sub>3</sub>O<sub>4</sub> composite electrodes while the formation of compact and rigid surface structures are observed for samples with increasing Fe<sub>3</sub>O<sub>4</sub> loading, as depicted in SEM photograph 1 (a) and 1 (b) show that the pitted and fragmented surfaces are more prominent for the pure AC and 4 wt. % Fe<sub>3</sub>O<sub>4</sub> composite electrodes while the formation of compact and rigid surface structures are observed for samples with increasing Fe<sub>3</sub>O<sub>4</sub> loading, as depicted in SEM photograph

1(e) and 1(f). The compact surface contour will retard the ion diffusion to the porous AC electrode, resulting in the loss of overall capacitance of the composite electrodes, as depicted in Table 2. The present findings seem to be consistent with other research study, which found that increasing the amount of ruthenium oxide would decrease the surface roughness of the composite electrodes and subsequently distort the specific capacitance (Kim & Popov 2002).

In order to elucidate the porosity property of the electrode, BET analysis has been conducted. The BET surface areas of the samples are summarized in Table 1. The BET analysis showed that the addition of 4 wt. % Fe<sub>2</sub>O<sub>4</sub> increases the specific surface area of pure AC electrode from 916 to 949 m<sup>2</sup>g<sup>-1</sup> and thus improves the capacitive performance improvement of the composite electrode effectively. This observation is in good accordance to research finding by Zhang and Chen (2008), which clearly indicate that the enhanced capacitance is mainly attributed to the high surface area of the core electrode material. On the other hand, further increase of Fe<sub>3</sub>O<sub>4</sub> compound to 60 wt. % causes the specific surface area of the composite electrode decreased to 800  $m^2g^{\mbox{-}1}$  or 15.7% surface area reduction. The specific surface area of pure  $Fe_3O_4$  (38 m<sup>2</sup>g <sup>1</sup>) is lower than that of the AC; when the  $\text{Fe}_3\text{O}_4$ 's content is increased, the specific surface area of the composite decreases. Fe<sub>3</sub>O<sub>4</sub> nanopowders probably gathered to form agglomerates within AC electrodes as shown in Figure 1(f). The crystal growth of Fe<sub>3</sub>O<sub>4</sub> reduces the ionic migration and thus causes the increase in diffusion path length. A fade in capacitance as a result of reduced surface area has been reported by several researchers in similar application (Hu & Chen 2004; Kim & Popov 2002).

#### ELECTROCHEMICAL CHARACTERIZATION

Cyclic Voltammogram (CV) Figures 2 and 3 show the CV plots for the Fe<sub>3</sub>O<sub>4</sub>/AC nanocomposite electrodes with different Fe<sub>3</sub>O<sub>4</sub> loading in 1 M Na<sub>2</sub>SO<sub>3</sub> and 1 M  $Na_2SO_4$  electrolytes, respectively, at a constant scan rate of 10 mV s<sup>-1</sup>. It can be seen that the voltammograms for the samples are not symmetrical along the current axes, especially in Na<sub>2</sub>SO<sub>3</sub> solution. The fact that the voltammograms do not show perfect rectangular features with a mirror image characteristic implies that there is a substantial pseudocapacitance contribution to the overall measured capacitance. In this study, it can be seen that the current response of the 4 wt. % Fe<sub>2</sub>O<sub>4</sub> electrodes in 1 M Na<sub>2</sub>SO<sub>2</sub> electrolyte is found to be greater than those of AC electrode and other composite electrodes, suggesting that highest specific capacitance is achieved for such composition. It is believed that the presence of  $Fe_2O_4$ nanoparticles in AC electrodes can provide favourable surface adsorption sites for sulphite anions which act as catalysts for subsequent redox and intercalation reactions. Interestingly, the types of electrolyte actually is playing a vital role in this context since the CV performance of the electrode material in  $Na_2SO_4$  solution is found to be



 $\begin{array}{l} \mbox{FiGURE 1. Scanning electron micrographs of different Fe}_{3}O_{4} \mbox{ loading in AC electrodes: (a) AC, (b) 4 wt. \% Fe}_{3}O_{4}, (c) 6 wt. \% Fe}_{3}O_{4}, (d) 8 wt. \% Fe}_{3}O_{4}, (e) 20 wt. \% Fe}_{3}O_{4}, (f) 60 wt. \% Fe}_{3}O_{4} \end{array}$ 

Electrode material	BET surface area (m <sup>2</sup> /g)		
СВ	1338.44		
AC	916.00		
4 wt. % Fe <sub>3</sub> O <sub>4</sub> /AC/CB	949.03		
8 wt. % Fe <sub>3</sub> O <sub>4</sub> /AC/CB	831.76		
60 wt. % Fe <sub>3</sub> O <sub>4</sub> /AC/CB	800.73		
Fe <sub>3</sub> O <sub>4</sub>	38.41		

TABLE 1. BET surface area of  $Fe_3O_4$ -AC composite electrodes

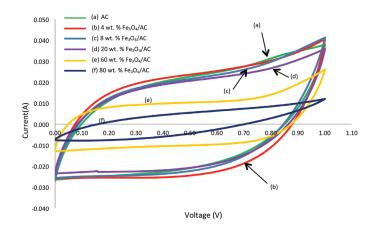


FIGURE 2. Cyclic voltammograms for different compositions of  $\text{Fe}_3\text{O}_4$ loaded in AC electrodes, scanned at 10 mV/s, in 1 M Na<sub>2</sub>SO<sub>3</sub>

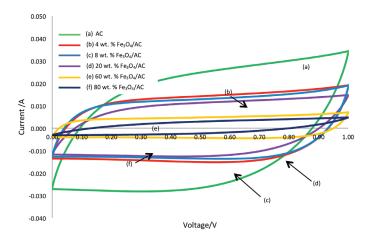


FIGURE 3. Cyclic voltammograms for different compositions of Fe<sub>3</sub>O<sub>4</sub> loaded in AC electrodes, scanned at 10 mV/s, in 1 M Na<sub>2</sub>SO<sub>4</sub>

distinctly different if compared with Na<sub>2</sub>SO<sub>2</sub> electrolyte. In fact, the AC electrode is found to exhibit the highest specific capacitance when 1 M Na<sub>2</sub>SO<sub>4</sub> solution is adopted as the electrolyte. Nevertheless, there is no obvious redox peaks could be observed for the AC electrodes in both solutions. An implication of these results suggests that the  $Fe_3O_4$  loaded AC composite electrodes and the pure AC electrode may have different storage mechanisms in sulphite and sulphate electrolytes, respectively. Wu et al. (2003) have characterized the capacitive behaviour of Fe<sub>3</sub>O<sub>4</sub>- carbon black electrodes in several aqueous electrolytes including Na<sub>2</sub>SO<sub>3</sub> and Na<sub>2</sub>SO<sub>4</sub>. The authors suggested that the significant capacitive performance in Na<sub>2</sub>SO<sub>3</sub> is attributed from the synergistic effect of both EDLC and the pseudocapacitance which involves successive reduction of the adsorbed sulphite anions. The capacitive current when  $Na_2SO_4$  is used as electrolyte on the other hand is due to the contribution from EDLC mechanism only. Figure 4 shows the specific capacitance of  $Fe_3O_4/AC$ composite electrodes as a function of Fe<sub>3</sub>O<sub>4</sub> loading, in 1

M Na<sub>2</sub>SO<sub>3</sub> and 1 M Na<sub>2</sub>SO<sub>4</sub>, respectively, at the scan rate of 10 mV/s between 0 and 1V.

Table 2 summarizes the effects of electrolyte and different Fe<sub>3</sub>O<sub>4</sub> loading on the specific capacitance of the electrodes. The enhancement of electrochemical storage properties of the AC electrode with a small amount of Fe<sub>3</sub>O<sub>4</sub> nanoparticles in Na<sub>2</sub>SO<sub>3</sub> is clearly observed, mainly attributed to the synergistic effect of the pseudocapacitance metal oxide with the double layer capacitance AC material. The presence of the  $Fe_2O_4$  nanoparticles could reduce the agglomeration and increase the surface area of AC material electrode. These results are also in good agreement with our observations in SEM (Figure 1) and BET analysis (Table 1). The large porosity of the structure with a higher specific surface area facilitates a better accessibility of the electrolyte inside the bulk of the electrodes during the electrochemical process. In addition to the increase of the effective surface area, a small amount of Fe<sub>3</sub>O<sub>4</sub> is also able to supply more electroactive sites for the adsorption/ de-adsorption of sulphite (SO32-) anions and promotes

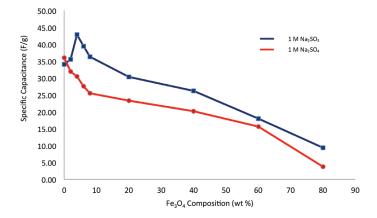


FIGURE 4. Specific capacitance of  $Fe_3O_4$ -AC composite electrodes as a function of  $Fe_3O_4$  loading, in 1 M Na<sub>2</sub>SO<sub>3</sub> and 1 M Na<sub>2</sub>SO<sub>4</sub> respectively, at the scan rate of 10 mV/s between 0 to 1V

TABLE 2. Effects of electrolyte and  $Fe_3O_4$  loading on specific capacitance of  $Fe_3O_4$ /AC composite electrodes

Electrode Material	Specific capacitance (F/g)			
	1 M Na <sub>2</sub> SO <sub>3</sub>	$1 \text{ M Na}_2 \text{SO}_4$		
AC	34.08	36.03		
2 wt. % Fe <sub>3</sub> O <sub>4</sub> /AC	35.52	31.96		
4 wt. % $Fe_3O_4/AC$	42.88	30.43		
6 wt. % $Fe_3O_4/AC$	39.45	27.55		
8 wt. % Fe <sub>3</sub> O <sub>4</sub> /AC	36.33	25.50		
20 wt. % Fe <sub>3</sub> O <sub>4</sub> /AC	30.33	23.30		
40 wt. % Fe <sub>3</sub> O <sub>4</sub> /AC	26.22	20.20		
60 wt. % $Fe_{3}O_{4}^{4}/AC$	18.00	15.60		
80 wt. % Fe <sub>3</sub> O <sub>4</sub> /AC	9.33	3.72		

subsequent redox reactions to occur effectively across the interface of electrode-electrolyte.

Nevertheless, as clearly indicated in Figure 4, the further increase in  $Fe_3O_4$  content within the range 20-60wt. % starts deteriorating the capacitive behaviour of the composite electrodes. The decrease in specific capacitance could be attributed to the low conductivity of  $Fe_3O_4$ nanoparticles in large amount, which in turn increase the equivalent series resistance (ESR) of the electrode. The specific capacitances of pure Fe<sub>3</sub>O<sub>4</sub> electrodes were found to be less than 9.4 and 4.0 F/g in Na<sub>2</sub>SO<sub>3</sub> and Na<sub>2</sub>SO<sub>4</sub> respectively, which clearly dictates its poor capacitive performance. Our observations are well complement with the similar effort by Wu et al. (2003). They reported that Fe<sub>2</sub>O<sub>4</sub> has a very poor conductivity and preliminary tests showed that it was necessary to combine the iron oxide with some conductive additives in order to significantly enhance its capacitive performance.

On top of low conductivity of  $Fe_3O_4$ , a fade in capacitance with increasing  $Fe_3O_4$  could be also due to the particle agglomeration, as shown in Figure 1(e) and 1(f). This phenomenon is similar to another investigation on the hydrous  $RuO_2$ -AC composite electrode, which a significant

distortion on the capacitive behaviour is observed for the electrode with more than 40 wt. % RuO<sub>2</sub> loading (Kim & Popov 2002). Since ion diffusion through the active materials controls the redox reactions of the electrode, the diffusive capability of these materials will decrease with increasing of the particle size. Under such circumstances, the ions thus require longer time to reach the entire electrodes. In addition, the agglomeration of the metal oxide nanoparticles will reduce the porosity of the electrode material, which in turn distorts the effective surface area that is available for the redox process. Similar trend has been observed by Dubal et al. (2012) for galvanostatically deposited Fe doped MnO<sub>2</sub> thin films. The low porosity of composite electrodes contributes to the poor diffusive capability since the large Fe<sub>3</sub>O<sub>4</sub> precipitates agglomerated on the AC surface and block most of the pores, preventing the ions to diffuse into the inner pores of the electrodes.

#### ELECTROCHEMICAL IMPEDANCE SPECTROSCOPY (EIS)

The EIS measurement was carried out on AC and  $\text{Fe}_3\text{O}_4$ /AC composite electrodes in the frequency range of 10<sup>4</sup> to 10<sup>-2</sup> Hz. Figure 5 shows the Nyquist plots of AC impedance of the composite electrodes with different

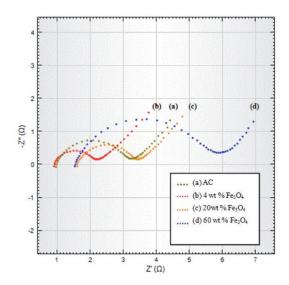


FIGURE 5. Nyquist plots obtained for composite electrodes with different Fe<sub>3</sub>O<sub>4</sub> loading, at 1 V over the frequency range of 100 kHz - 100 mHz in 1 M Na<sub>3</sub>SO<sub>3</sub>

 $Fe_3O_4$  compositions in Na<sub>2</sub>SO<sub>3</sub>. It can be seen that all the impedances consist of an arc and a slanted line at low frequency. In the high frequency region, the intercept of the semicircle on the real axis of the Nyquist spectrum represents the solution resistance (R<sub>sol</sub>) which provides the ohmic resistance of the electrolyte and the internal resistance of the electrode materials. The semicircles in the high- and mid-frequency regions are attributed to the charge transfer resistance at the interface of the Fe<sub>3</sub>O<sub>4</sub>-Fe<sub>3</sub>O<sub>4</sub>, AC-Fe<sub>3</sub>O<sub>4</sub>, AC-AC composite and the Na<sub>2</sub>SO<sub>3</sub> electrolyte. The electron transfer which occurs in these regions during the charge/discharge processes is modelled by an interfacial charge transfer resistance  $(R_{r})$  or polarisation resistance  $(R_{r})$ . After the semicircle, the Nyquist spectrum shows a long tail in the lowfrequency region pertaining to the Warburg resistance or the diffusion of ions into the bulk of the electrode. As shown in Figure 5, it is apparent that the sizes (diameter) of the Nyquist semicircles are varied for the electrodes with different Fe<sub>3</sub>O<sub>4</sub> composition. When Na<sub>2</sub>SO<sub>3</sub> is used as an electrolyte, the  $R_{ct}$  for the electrodes with different Fe<sub>3</sub>O<sub>4</sub> composition is found to decrease in the order of 4.28  $\Omega$  (60 wt. %), 1.87  $\Omega$  (20 wt. %) and 1.37  $\Omega$  (4 wt. %). On the other hand, the R<sub>ct</sub> of the pure AC electrode is found to be 2.30  $\Omega$ . From these results, it is clear that the 4 wt. % Fe<sub>3</sub>O<sub>4</sub>/AC composite electrode exhibits the lowest R<sub>ct</sub> value, consistent with the smallest size (diameter) of the Nyquist semicircle within the high frequency region. A reduction in the value of R<sub>et</sub> implies that the surface modification by addition of fairly small amount of Fe<sub>3</sub>O<sub>4</sub> can enhance the conductivity of the composite electrode and hence improves its capacitive performance, in agreement with the results obtained from CV analysis.

The fitting of the impedance expressions obtained from the equivalent circuit to the experimental Nyquist

spectra was performed using the NOVA equivalent circuit editor in order to determine the most representative physical processes that taking place in the systems under investigation (Figure 6). In the present study, the double layer capacitor in EIS experiments often behaves like constant-phase element (CPE) as a result of imperfection on the real electrochemical cell including surface roughness, leaky capacitor and non-uniform current distribution (Jovic 2012). The negative phase at high frequencies shows the effect of inductance error (L) on the impedance spectra of an electrochemical cell attributed to the improper length of lead and wiring configuration (MacDonald & Barsoukov 2005).

Table 3 summarises the equivalent circuit parameters obtained from the fitting results in 1 M Na<sub>2</sub>SO<sub>3</sub>. From the table, it is found that 4 wt. % Fe<sub>3</sub>O<sub>4</sub> in the composite electrode shows relatively lower  $R_{\Omega}$  (0.83  $\Omega$ ) and  $R_{ct}$ (1.37  $\Omega$ ) in comparison to R<sub>o</sub> (0.90  $\Omega$ ) and R<sub>ct</sub> (2.30  $\Omega$ ) of pure AC electrode. A reduced value of R<sub>CT</sub> implies that the surface modification with very small amount of  $Fe_3O_4$  enhances the conductivity of the composite electrodes. Additionally, Warburg impedance (W) of 4 wt. % Fe<sub>3</sub>O<sub>4</sub> shows a much lower value compared to pure AC electrode, which corresponds to more vertical line leaning to the imaginary axis at a low frequency region. Similar EIS investigations were reported in other studies as well (Masarapu et al. 2009; Wang et al. 2012) showing that enhanced capacitive performance is related to lower Warburg impedance values. These findings show that 4 wt. % Fe<sub>2</sub>O<sub>4</sub> demonstrates more ideal capacitive behaviour due to more facile electrolyte diffusion to the composite surface.

In general, porous electrodes generate an exponent value of n = 0.5 while rough electrode possesses n value between 0.5 and 1 (Masarapu et al. 2009). In this

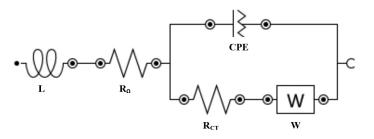


FIGURE 6. The equivalent circuit used to fit the Nyquist spectra using NOVA equivalent circuit editor

TABLE 3. Simulated values of L.  $R_{\Omega}$ ,  $R_{CT}$ , CPE, W, n from the equivalent circuit in Figure 6

Electrode composition	Inductance, L (nH)	Solution resistance , $R_{\Omega}$ ( $\Omega$ )	Charge transfer resistance, $R_{CT}$ ( $\Omega$ )	Constant-phase element, CPE (µMho)	Warburg element, W (mMho)	Exponent (n)
AC	282	0.90	2.30	475	685	0.723
4 wt. % $Fe_{3}O_{4}$	230	0.83	1.37	772	487	0.695
20 wt. % Fe <sub>3</sub> O <sub>4</sub>	235	1.55	1.87	583	622	0.710
60 wt. $\% \text{ Fe}_{3}\text{O}_{4}$	274	1.48	4.28	452	519	0.727

case, it can be observed that the values of *n* increase with increasing  $\text{Fe}_3\text{O}_4$  loading. The higher value of the exponent in 20 and 60 wt. %  $\text{Fe}_3\text{O}_4$  shows that these composite electrodes are behaving as typical non-porous electrodes in which the ion diffusion is taking place only on the surface of the composite electrodes. These findings are consistent with SEM micrographs (Figure 1) and BET surface area measurement (Table 1) on composite electrodes with different  $\text{Fe}_3\text{O}_4$  loading. The non-porous nature limits the number of ions that diffuse into the bulk of electrodes and thus yields lower specific capacitance compared to the porous electrodes.

### CONCLUSION

The capacitive performance of Fe<sub>3</sub>O<sub>4</sub> -AC nanocomposite electrode materials in Na<sub>2</sub>SO<sub>3</sub> and Na<sub>2</sub>SO<sub>4</sub> has been characterized. The highest specific capacitance of 43 F/g is achieved with a relatively low  $Fe_3O_4$  loading (4 wt. %) in 1 M Na<sub>2</sub>SO<sub>3</sub>. However, bulk increasing of the Fe<sub>3</sub>O<sub>4</sub> content is found to distort the capacitive performance and deteriorate the specific surface area of the electrode, mainly due to the aggregation of the Fe<sub>3</sub>O<sub>4</sub> particles within the composite. Additionally, the Fe<sub>3</sub>O<sub>4</sub>-AC nanocomposite electrode in Na<sub>2</sub>SO<sub>3</sub> electrolyte exhibits a better charge storage performance if compared with Na<sub>2</sub>SO<sub>4</sub> solution. It is believed that Fe<sub>2</sub>O<sub>4</sub> nanoparticles can provide favourable surface adsorption sites for sulphite (SO<sub>2</sub><sup>2-</sup>) anions during the redox and intercalation reactions. The improvement of the capacitive performance of the 4 wt. % Fe<sub>2</sub>O<sub>4</sub> -AC composite electrode is believed attributed to the contribution of synergistic effect of EDLC and pseudocapacitance charge storage mechanism, as well as the enhancement on the specific surface areas of the electrode.

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